

Synthesis and Structural Characterization of Two New Rare-Earth Manganese Germanates: $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ and $\text{GdMnGe}_2\text{O}_7$

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Crystals of $\text{Ce}^{+4}\text{Mn}_2^{+2}\text{Ge}_4\text{O}_{12}$ and $\text{Gd}^{+3}\text{Mn}^{+3}\text{Ge}_2\text{O}_7$ were obtained during the exploration of the quaternary Ln-Mn-Ge-O system. Their structures have been determined by single-crystal X-ray diffraction: $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ crystallizes in the tetragonal space group P4/nbm with $a = 9.816(1) \text{ \AA}$, $c = 4.888(1) \text{ \AA}$, $Z = 2$, and $R = 0.026$ while $\text{GdMnGe}_2\text{O}_7$ crystallizes in the orthorhombic space group A222 with $a = 4.735(1) \text{ \AA}$, $b = 7.839(2) \text{ \AA}$, $c = 13.500(3) \text{ \AA}$, $Z = 4$, and $R = 0.048$. Both frameworks are built up from the same polyhedra, i.e., LnO_8 antiprisms, MnO_6 octahedra, and GeO_4 tetrahedra. $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ is isotypic with $\text{SrNa}_2\text{P}_4\text{O}_{12}$; the GeO_4 tetrahedra are linked into Ge_4O_{12} rings located between layers of CeO_8 and MnO_6 polyhedra sharing edges. On the other hand, $\text{GdMnGe}_2\text{O}_7$ exhibits a novel quasi-one-dimensional lattice which might be ascribed to the Jahn–Teller nature of the Mn^{+3} ion. The structure consists of chains of MnO_6 octahedra alternating with parallel chains of GdO_8 antiprisms forming layers linked together through Ge_2O_7 groups. A structural comparison with fluorite-related compounds as well as $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ will be discussed. © 1999 Academic Press

INTRODUCTION

The association of a transition metal element with a covalent element such as phosphorus, silicon, or germanium in oxides generates varied opened structures. Only a few of such compounds with manganese are reported in the literature although the peculiar behaviour of this element should make it possible to isolate a great number of new materials. Indeed, the coordination figures formed by oxygen atoms around manganese atoms are manifold and as a rule they exhibit four to eight ligands (1). Moreover, manganese can take various oxidation states ranging from II to VII. Finally, the Jahn–Teller nature of the Mn^{3+} (d^4) ion might allow a unique framework to form.

Recently, our investigation of the system Ln-Mn-Si/Ge-O has produced interesting examples of oxo-compounds,

$\text{La}_4\text{Mn}_5\text{Si}_4\text{O}_{22}$ (2) and PrMnOGeO_4 (3), containing rutile-like MnO_2 layers or infinite chains of MnO_6 octahedra, respectively. The quasi-low-dimensional character of these materials is enhanced by the presence of large rare-earth cations. Furthermore, $\text{La}_4\text{Mn}_5\text{Si}_4\text{O}_{22}$ displays the unusual characteristic of three oxidation states of manganese: +2, +3, and +4.

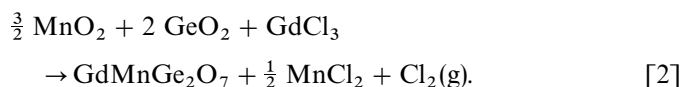
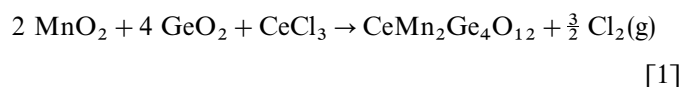
Over the years, the study of oxo-compounds characterized by quasi-one-dimensional chain or two-dimensional layer transition metal oxide structures which are isolated by closed-shell groups like silicate or phosphate anions, have received much attention. Besides the structural aspect, these compounds have shown to be of interest to the study of the behaviour of delocalized electrons in a confined lattice (4).

The present paper deals with the syntheses and the structural analyses of two new rare-earth manganese germanates: $\text{Ce}^{+4}\text{Mn}_2^{+2}\text{Ge}_4\text{O}_{12}$ and $\text{Gd}^{+3}\text{Mn}^{+3}\text{Ge}_2\text{O}_7$. $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ is isostructural with the anhydrous cyclotetraphosphate $\text{SrNa}_2\text{P}_4\text{O}_{12}$ (5), whereas $\text{GdMnGe}_2\text{O}_7$ exhibits a novel quasi-one-dimensional lattice formed by chains of MnO_6 octahedra which has not previously been observed in the large AMX_2O_7 compounds family (6, 7). A structural comparison with fluorite-related compounds (8) as well as $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ will be briefly discussed.

EXPERIMENTAL SECTION

Synthesis and Characterization Techniques

$\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ and $\text{GdMnGe}_2\text{O}_7$ were prepared by a redox reaction involving LnCl_3 (Ln: Ce, Gd), MnO_2 and GeO_2 :



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The reaction mixtures were placed in evacuated quartz tubes, then heated for 10 days at 700°C and slow-cooled at 10°C/h to room temperature. In order to avoid explosions due to chlorine gas emission, it is important to choose an adequate volume of the tube. The resulting product of reaction [2] was stirred in alcohol to dissolve the MnCl₂ matrix.

Both crystals of CeMn₂Ge₄O₁₂ and GdMnGe₂O₇ are light red needles.

The compositions were confirmed by X-ray microanalyses using a JEOL JSM-35C scanning electron microscope equipped with a Tracor TN 5500 micro Z system. Mn₇SiO₁₂ was a common impurity resulting from attack on the silica tube (9). Neither Si nor Cl impurity was detected in the crystals of CeMn₂Ge₄O₁₂ and GdMnGe₂O₇, however.

Structure Determination

Initially, the single crystals used for data collection were investigated by oscillation and Weissenberg-type film methods. It is noteworthy that GdMnGe₂O₇ crystallizes in a noncentrosymmetric space group A222; attempts to refine the structure in the corresponding centrosymmetric space group Amm (No. 65) failed. Data collection was performed on an Enraf-Nonius CAD4 diffractometer (see Table 1 for details). Both crystal structures were solved by direct and Fourier methods. They were refined by full-matrix least-squares techniques using the SHELXTL-PLUS program (10). Data were corrected for Lorentz polarization, absorption (11), and secondary extinction. The atomic scattering factors were those of Cromer and Waber (12) and the corrections for anomalous dispersions were from Cromer and Waber (13). Tables 2 and 4 list the final positional and thermal parameters.

RESULTS AND DISCUSSION

CeMn₂Ge₄O₁₂

The structure determination of CeMn₂Ge₄O₁₂ shows that it is isotypic with the anhydrous cyclotetraphosphate SrNa₂P₄O₁₂ (5). Compounds with the general formula AB₂X₄O₁₂ crystallize in three related structure types (5, 15): M^I₂M^{II}P₄O₁₂ (M^I = K, NH₄, Rb, Tl and M^{II} = Pb, Sr, Ba), Na₂M^{II}P₄O₁₂ (M^{II} = Ca, Sr) and Ca₂ZrSi₄O₁₂. While many studies have been devoted to cyclophosphates and cyclosilicates, little data on the crystal structures of cyclo-germanates have been reported (14).

The structure of CeMn₂Ge₄O₁₂ (Fig. 1) consists of (001) layers of CeO₈ antiprisms and distorted MnO₆ octahedra sharing edges (Fig. 2a), linked together via O-Ge-O bridges of the Ge₄O₁₂ rings; each MnO₆ octahedron shares all its corners with four different Ge₄O₁₂ rings (Fig. 2b).

Selected interatomic distances and bond angles are given in Table 3. The Ge₄O₁₂ rings show distances and angles

TABLE 1
Crystal Data, X-Ray Data Collection, and Structure Determination

Empirical formula	CeMn ₂ Ge ₄ O ₁₂	GdMnGe ₂ O ₇
Formula weight (g/mol)	732.42	469.40
Crystal system	Tetragonal	Orthorhombic
Space group	P4/nbm (No. 125)	A222 (No. 21)
Lattice parameters:		
a (Å)	9.816(1)	4.735(1)
b (Å)	9.816(1)	7.839(2)
c (Å)	4.888(1)	13.500(3)
V (Å ³)	470.98	501.09
Z	2	4
Temperature (K)	294	294
Color and habit	Light red needle	Light red needle
Crystal dimensions (mm)	0.05 × 0.03 × 0.03	0.05 × 0.03 × 0.03
D _{calc} (g/cm ³)	5.16	6.20
Diffractometer	Enraf-Nonius CAD4	Enraf-Nonius CAD4
Radiation	MoKα (λ = 0.71073 Å)	MoKα (λ = 0.71073 Å)
μ _(MoKα) (cm ⁻¹)	208.84	142.98
Scan type	ω	ω
θ range (°)	1.0–50.0	1.5–50.0
Scale factor	1.86877	0.74322
Quadrants measured (hkl)	(-27, -1, -1) to (27, 11, 9)	(-27, 0, -1) to (27, 11, 19)
No. of reflections measured	1572	2226
No. observations (I > 3.00σ(I))	404	1020
No. variables	27	54
Corrections:		
Lorentz polarization		
Absorption		
Residuals ^a : R, R _w	0.026, 0.028	0.048, 0.030
Final diff. Fourier (e-/Å ³)	1.94	3.90

$$^a R = \sum ||F_o| - |F_c|| / \sum |F_o| \quad \text{and} \quad R_w = \sum (||F_o| - |F_c|| \cdot \sqrt{w}) / \sum (|F_o| \cdot \sqrt{w}); \quad w = 1/\sigma^2(F_o).$$

similar to those found in other cyclotetragermanates (14(b)). The MnO₆ octahedra are elongated with two apical oxygen atoms at distances of 2.412 Å, longer than the four equatorial ones, 2.121 Å, leading to an averaged Mn-O distance of 2.218 Å expected for Mn²⁺. This distortion cannot be ascribed to Jahn-Teller effects for high spin Mn²⁺ has a nondegenerate ground state. It may be noticed that sodium atoms in SrNa₂P₄O₁₂ exhibit a similar distorted octahedral coordination. Such distorted environments around Mn²⁺ ions have already been observed in highly rigid frameworks of minerals (16). Bond valence sum calculations using Brese's tables (17) corroborate this oxidation state assignment; Mn gives 1.78 valence units (v.u.). Therefore, the oxidation state of cerium atoms must be 4+ in order to maintain the electroneutrality; the calculated valence is +3.58 v.u..

GdMnGe₂O₇

GdMnGe₂O₇ exhibits a novel quasi-one-dimensional structure formed by chains of MnO₆ octahedra together

TABLE 2
Positional and Equivalent Displacement Parameters for
 $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$

Atom	Wyckoff	x	y	z	U_{eq}
Ce	2b	1/4	1/4	1/2	0.0053(1)
Mn	4f	0	0	1/2	0.0102(2)
Ge	8k	0.52379(6)	1/4	0	0.0064(1)
O1	8m	-0.3703(9)	0.3703(9)	0.1714(3)	0.0086(9)
O2	16n	0.1668(3)	0.0643(4)	0.2565(8)	0.0081(8)

with Ge_2O_7 groups and GdO_8 antiprisms. During the past years, a large number of compounds of stoichiometry AMX_2O_7 have been isolated (6, 7). Their mixed frameworks consist of AO_m polyhedra (A: alkali or alkaline-earth metal; $m = 4$ to 10), MO_n polyhedra ($M: \text{M}^{2+}$ or M^{3+} transition metal; $n = 4$ to 6) and X_2O_7 groups ($X: \text{P}, \text{Si}, \text{Ge}, \text{As}$). Those built up from MO_6 octahedra incorporate either individual MO_6 octahedra or edge-sharing M_2O_{10} bioctahedral units. To the best of our knowledge, none of them displays chains of MnO_6 octahedra.

Figure 3 shows the structure of $\text{GdMnGe}_2\text{O}_7$ viewed along the chains of corner-sharing MnO_6 octahedra. These undulating chains alternate with chains of edge-sharing GdO_8 antiprisms forming layers parallel to (100) (Fig. 4a). In the a direction, the MnO_6 chains are cross-linked via O-Ge-O bridges of the Ge_2O_7 groups resulting other layers that can be described as containing $\text{MnGe}_2\text{O}_{11}$ units (Fig. 4b). Each unit, in which one Ge_2O_7 group shares two of its apices with the same MnO_6 octahedron, is linked to

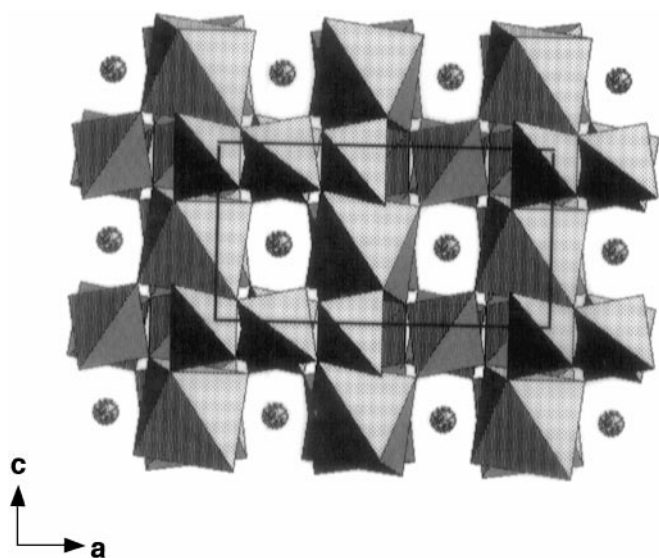


FIG. 1. Structure of $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ in the ac -plane. Small filled circles represent cerium atoms.

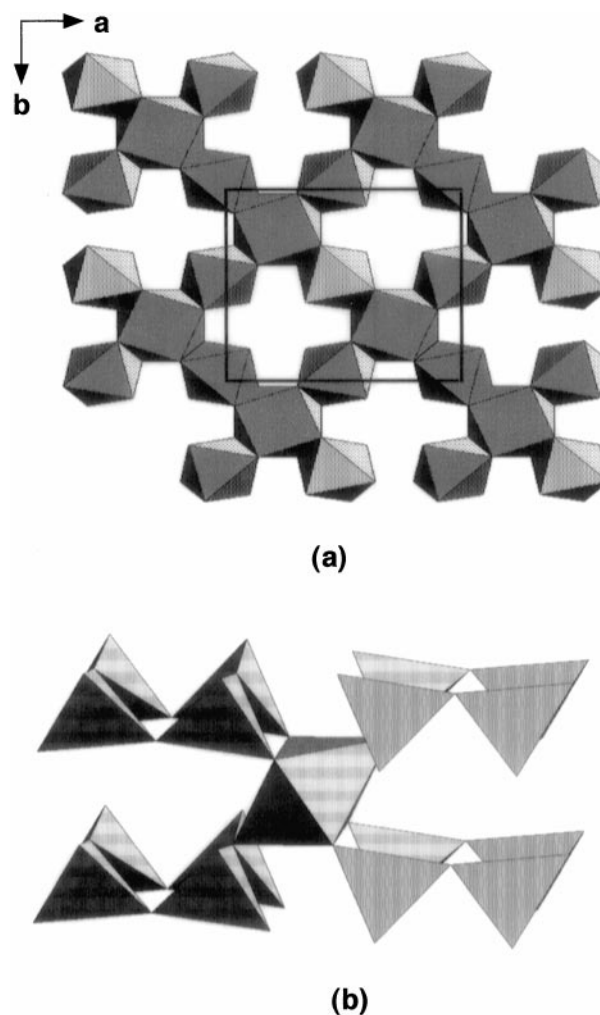


FIG. 2. (a) (001) layers in $\text{CeMn}_2\text{Ge}_4\text{O}_{12}$ formed by MnO_6 octahedra and CeO_8 antiprisms sharing edges; (b) these layers are connected in the c direction by Ge_4O_{12} rings.

five other units lying in the same plane. Such an arrangement of MO_6 and X_2O_7 polyhedra has never been shown so far. On the other hand, $\text{GdMnGe}_2\text{O}_7$ possesses the same arrangement of MO_6 - LnO_8 slabs as in the fluorite-related phases Ln_3MO_7 (Ln: rare-earth, Y or Sc and $M: \text{Nb}, \text{Sb},$ or Ta) (8), as well as in the pyrochlore and weberite compounds (18). They differ basically only in the interslab cations though the LnO_8 polyhedra are nearly cubes in the latter compounds.

Table 5 lists selected bond lengths and angles for GdO_8 , MnO_6 , and Ge_2O_7 polyhedra. Bond valence sum calculations for Mn and Ge sites are consistent with their formal oxidation states at +2.99 and +3.81 v.u., respectively. The distorted octahedral coordination of the Mn site with Mn-O axial bond lengths of 2.464 Å much longer than the equatorial ones, 1.932–1.883 Å, is suitable to the Jahn-Teller nature of the Mn^{3+} ion. The MnO_6 octahedra are linked

TABLE 3
Important Bond Lengths (Å) and Angles (°) in CeMn₂Ge₄O₁₂

Ce	O2 × 8	2.325(4)					
Mn	O1 × 2	2.412(4)	O1	Mn	O1	180.0	
	O2 × 4	2.121(4)	O2	Mn	O2	180.0 × 2	81.7(2) × 2
						98.3(2) × 2	
			O1	Mn	O2	97.1(2) × 4	82.9(2) × 4
Ge	O1 × 2	1.783(6)	O1	Ge	O1	108.7(1)	
	O2 × 2	1.728(4)	O2	Ge	O2	120.0(2)	
			O1	Ge	O2	105.3(2) × 2	
						108.6(2) × 2	
			Ge	O1	Ge	123.5(3)	

through the mutual sharing of the oxygen atom O1. It is worth pointing out the high coordination number of this oxygen atom which is both the bridging oxygen atom of MnO₆ chains and Ge₂O₇ groups. Bond valence sum calculations show a weak contribution of the long Mn-O1 bonds, $\nu_{\text{Mn-O1}} = +0.15$ v.u., allowing O1 to be shared with the Ge₂O₇ groups, $\nu_{\text{Ge-O1}} = +0.86$ v.u. The sum is in good agreement with the formal oxidation state of oxygen anion: -2.02 v.u. The zigzag configuration of MnO₆ chains is also rather uncommon: the long Mn-O1 bond results in a long Mn—Mn separation distance, 3.921 Å, with a very bent Mn-O1-Mn bond angle 105.5°. From these observations, it is inferred that only the Jahn–Teller nature of the Mn³⁺ cation would make it possible for such a constrained framework to be stabilized.

The environment of the rare-earth cation is an antiprism GdO₈ with Gd-O bond lengths ranging from 2.341 Å to 2.428 Å. The calculated valences are +3.49 for Gd1 and +3.39 for Gd2. The GeO₄ tetrahedra forming the Ge₂O₇ groups are distorted with Ge-O distances varying from 1.724 Å to 1.804 Å and O-Ge-O bond angles from 99.4 to 120.3°; this is due to the complex bond interaction occurring at each oxygen anion.

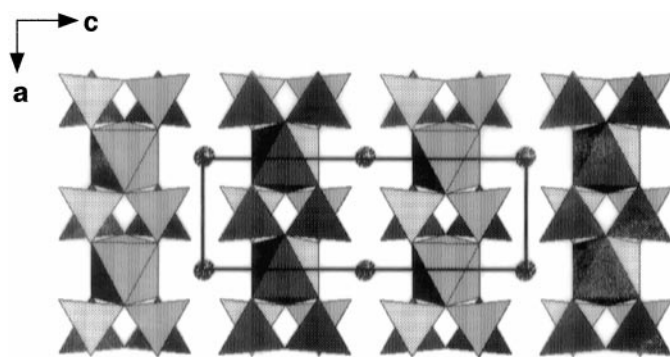


FIG. 3. Structure of GdMnGe₂O₇ in the *ac*-plane. Small filled circles represent gadolinium atoms.

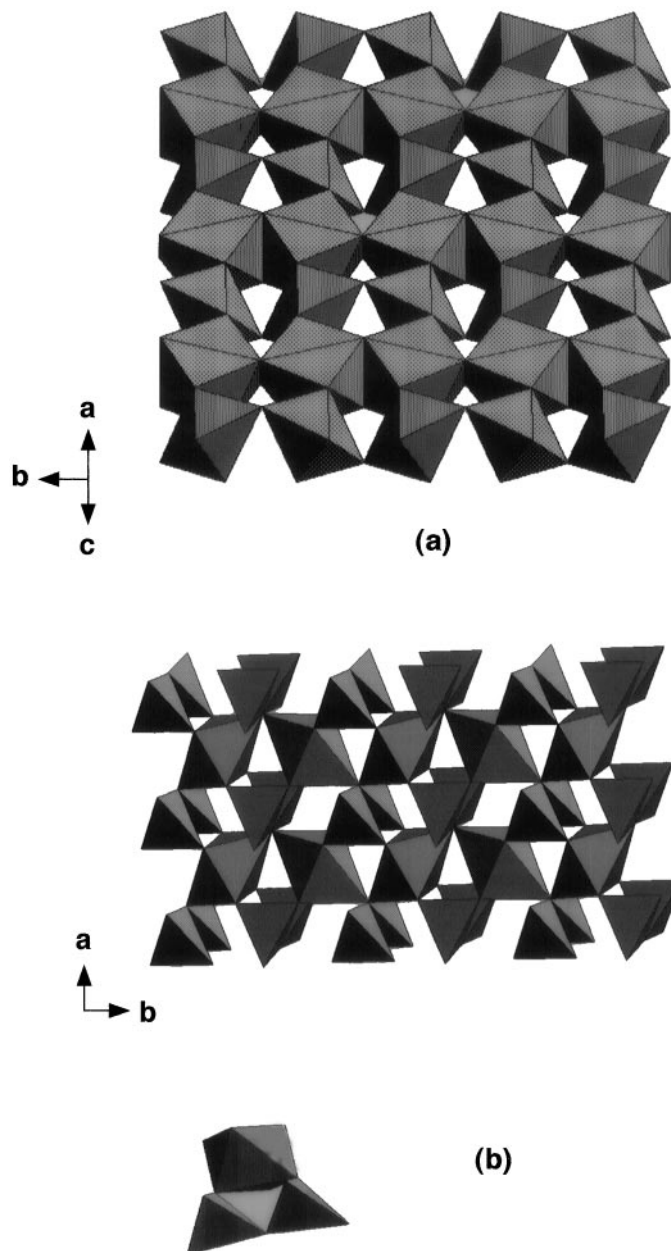


FIG. 4. (a) (100) layers in GdMnGe₂O₇ formed by alternating chains of MnO₆ octahedra and GdO₈ antiprisms. (b) The chains of MnO₆ octahedra are linked in the *a* direction through Ge₂O₇ groups resulting other layers parallel to (001) that can be described as containing MnGe₂O₁₁ units.

An interesting structural correlation can be drawn between GdMnGe₂O₇ and CeMn₂Ge₄O₁₂. Both structures are built up from the same polyhedra and the chemical formula of GdMnGe₂O₇ can be rewritten Gd₂Mn₂Ge₄O₁₄. Then, the two compounds differ from each other by two oxygen anions and a rare-earth cation whose charges are balanced by different electropositive cations, Ce⁴⁺ versus

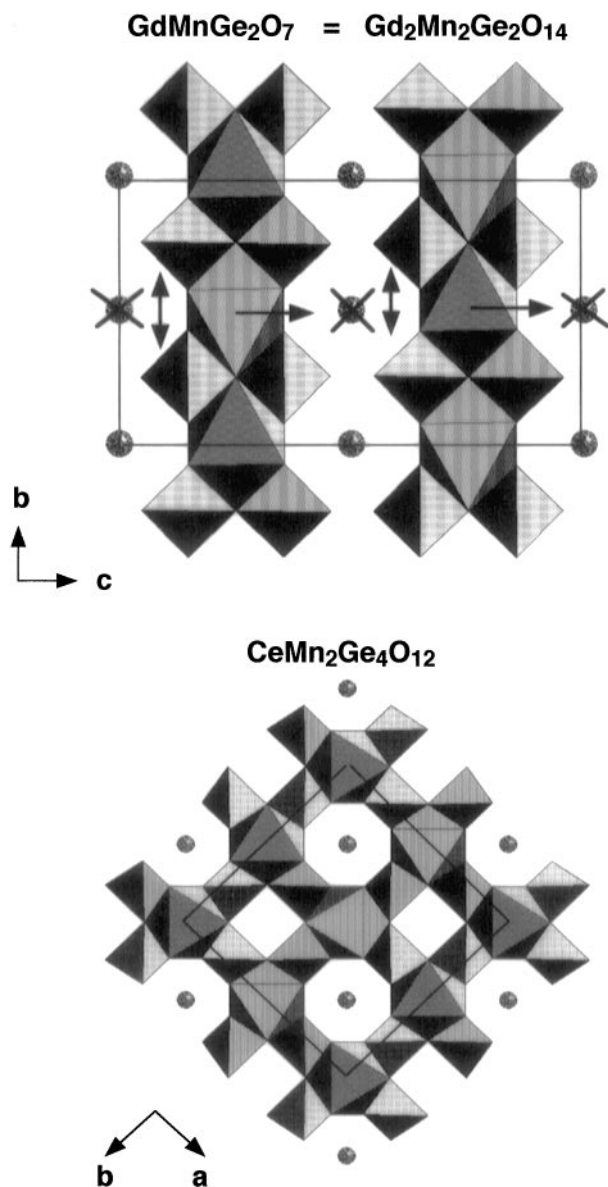


FIG. 5. Structural relationship between GdMnGe₂O₇ and CeMn₂Ge₄O₁₂. Filled circles marked with a cross represent rare-earth atoms in GdMnGe₂O₇ that are replaced by Mn atoms in CeMn₂Ge₄O₁₂.

TABLE 4
Positional and Equivalent Displacement Parameters for GdMnGe₂O₇

Atom	Wyckoff	x	y	z	U_{eq}
Gd1	2a	0	0	0	0.0029(3)
Gd2	2b	0	1/2	0	0.0071(4)
Mn	4e	0	0	0.7536(3)	0.0050(2)
Ge	8l	-0.5186(2)	-0.2488(4)	-0.13708(4)	0.0053(3)
O1	4k	-0.3151(9)	-3/4	-3/4	0.007(1)
O2	8l	0.238(2)	-0.425(1)	-0.1457(8)	0.008(1)
O3	8l	-0.2626(7)	0.246(2)	0.0462(3)	0.008(1)
O4	8l	0.244(1)	0.080(1)	0.1466(8)	0.005(1)

TABLE 5
Important Bond Lengths (Å) and Angles (°) in GdMnGe₂O₇

Gd1	O3 × 4	2.379(14)	O1	Mn	O1	177.7(2)			
	O4 × 4	2.377(10)				O2	82.8(3) × 2		
Gd2	O2 × 4	2.341(10)	O2	Mn	O2	95.5(3) × 2			
	O3 × 4	2.428(15)				O4	96.8(3)		
Mn	O1 × 2	2.464(3)	O4	Mn	O4	96.9(3)			
	O2 × 2	1.932(10)				O4	84.8(3) × 2		
	O4 × 2	1.883(10)				O2	82.1(6)		
	Mn × 2	3.921(4)				O4	Mn	O4	176.7(5) × 2
									O4
Ge	O1	1.803(3)	O4	Mn	O4	88.6(6)			
	O2	1.804(11)				Mn	O1	Mn	105.5(2)
	O3	1.724(4)				O1	Ge	O2	106.5(3)
	O4	1.740(11)				O2	Ge	O3	103.1(2)
									O4
	O3	Ge				O4	Ge	O4	99.4(5)
									O4
	Ge	O1				Ge	115.4(3)		

Gd³⁺ and Mn²⁺ versus Mn³⁺. As can be seen in Fig. 5, one could imagine the change from Gd₂Mn₂Ge₄O₁₄ to CeMn₂Ge₄O₁₂ in three steps: first, half the rare-earth sites are removed, then, half the MnO₆ octahedra are placed on these free sites, and finally, the Ge₂O₇ pairs facing each other are fused to give Ge₄O₁₂ cycles.

REFERENCES

- E. F. Bertaut, G. Buisson, A. Durif, J. Mareschal, M. C. Montmory, and S. Quezel-Ambrunaz, *Colloq. Int. CNRS Soc. Chim.* **5**, 1132 (1965).
- C. Gueho, D. Giaquinta, J. L. Mansot, T. Ebel, and P. Palvadeau, *Chem. Mater.* **7**, 486 (1995).
- C. Gueho, D. Giaquinta, P. Palvadeau, and J. Rouxel, *J. Solid State Chem.* **7**, 120 (1995).
- a. J. Rouxel, "Crystal Chemistry and Properties of Materials with Quasi-One-Dimensional Structures," Vol. 291, p. 263. Reidel, Dordrecht, 1986.
- b. R. C. Hausalter, M. E. Wang, J. Thompson, J. Zubieta, and C. J. O'Connor, *J. Solid State Chem.* **109**, 259 (1994).
- c. S. Wang and S. J. Hwu, *J. Am. Chem. Soc.* **114**, 6920 (1992).
- d. S. Wang, S.-J. Hwu, J. A. Paradis, and J. Whangbo, *J. Am. Chem. Soc.* **117**, 5515 (1995).
- M. T. Averbuch-Pouchot and A. Durif, *Acta Crystallogr. Sect. C* **39**, 811 (1983).
- a. E. Dvoncova and K. H. Lii, *J. Solid State Chem.* **105**, 279 (1993).
- b. L. Benhamada, A. Grandin, M. M. Borel, A. Leclaire, and B. Raveau, *Acta Crystallogr. Sect. C* **47**, 424 (1991).
- c. K. H. Lii, Y. P. Wang, Y. B. Chen, and S. L. Wang, *J. Solid State Chem.* **86**, 143 (1990).

- 6d. S. L. Wang, P. C. Wang, and Y. P. Nieh, *J. Appl. Crystallogr.* **23**, 520 (1990).
- 6e. Y. P. Wang, K. H. Lii, and S. L. Wang, *Acta Crystallogr. Sect. C* **45**, 673 (1989).
- 6f. Y. P. Wang and K. H. Lii, *Acta Crystallogr. Sect. C* **45**, 1210 (1989).
- 6g. D. Riou, A. Leclaire, A. Grandin, and B. Raveau, *Acta Crystallogr. Sect. C* **45**, 989 (1989).
- 6h. J. J. Chen, C. C. Wang, and K. H. Lii, *Acta Crystallogr. Sect. C* **45**, 673 (1989).
- 6i. A. Leclaire, M. M. Borel, A. Grandin, and B. Raveau, *J. Solid State Chem.* **78**, 220 (1989).
- 7a. R. D. Adams, R. Layland, and C. Payen, *Polyhedron*, **23–24**, 3473 (1995).
- 7b. D. Riou and B. Raveau, *Acta Cryst. C* **47**, 1708 (1991).
- 7c. A. Moquine, A. Boukhari, and E. M. Holt, *Acta Cryst. C* **47**, 2294 (1991).
- 7d. D. Riou, H. Leligny, C. Pham, P. Labbe, and B. Raveau, *Acta Cryst. B* **47**, 608 (1991).
- 7e. D. Riou and M. Goreaud, *Acta Cryst. C* **46**, 1191 (1990).
8. H. J. Rossel, *J. Solid State Chem.* **27**, 115 (1979).
9. P. Euzen, M. Queignec, P. Palvadeau, J. P. Venien, *Mat. Res. Bull.* **26**, 841 (1991).
10. *SHELXTL PLUS 4.0*, Siemens Analytical X-Ray Instruments, Madison, WI, 1989.
11. A. C. North, D. C. Phillips, and F. S. Matthews, *Acta Crystallogr. A* **24**, 351 (1968).
12. D. T. Cromer and J. T. Waber, in "International Tables for X-Ray Crystallography," Vol. 2.2A. Kynoch, Birmingham, UK, 1974.
13. D. T. Cromer and J. T. Waber, in "International Tables for X-Ray Crystallography," Vol. 2.3.1. Kynoch, Birmingham, UK, 1974.
- 14a. K. K. Palkina, S. I. Maksimova, V. G. Kuznetsov, and N. N. Chudinova, *Sov. Phys. Dokl.* **24**, 243 (1979).
- 14b. C. Pagnoux, A. Verbaere, Y. Kanno, Y. Piffard, and M. Tournoux, *J. Solid State Chem.* **99**, 173 (1992).
- 14c. A. Y. Barkov, Y. P. Men'Shikov, V. D. Begizov, and A. I. Lednev, *Eur. J. Mineral.* **8**, 311 (1996).
- 15a. C. Caverio-Ghersa and A. Durif, *J. Appl. Cryst.* **8**, 562 (1975).
- 15b. S. Colin, B. Dupre, G. Venturini, B. Malaman, and C. Gleitzer, *J. Solid State Chem.* **102**, 242 (1993).
- 15c. M. T. Averbuch-Pouchot, *Eur. J. Solid State Inorg. Chem.* **33**, 15 (1996).
16. P. B. Moore and T. Araki, *Am. Mineral.* **64**, 390 (1979).
17. N. E. Brese and M. O'Keeffe, *Acta Cryst. B* **47**, 192 (1991).
- 18a. A. Byström, *Ark. Kem. Min. Geol. A* **18**(10), (1944).
- 18b. A. Byström, *Ark. Kem. Min. Geol. A* **18**(21), (1944).